

Key-note report

**Key Technical Advances for
Optimizing Bulk Nanostructured Metals for Applications**

Terry C. Lowe

Los Alamos National Laboratory, Los Alamos, NM 87545, USA

tlowe@metallicum.com

Nanomaterials have been at the forefront of nanoscale research and subsequent product commercialization since the expansion of nanotechnology in the 1990s. Excluding the semiconductor industry, in 2007 nearly 90% of nanotechnology products in the U.S. were nanomaterials. Nanoparticles, nanofibers, nanotubes, nanolayers, and other nanoscale structures have been incorporated into products in U.S markets with a total value that now exceeds \$87 billion.

Bulk nanostructured metals and alloys are positioned to emerge as a significant portion of the nanomaterials sector, but only as we further understand three particular nanometal characteristics: the effects of nanoscale inhomogeneity on macroscale properties, the long term stability of nanostructures, and environmental interactions of nanostructured surfaces. This work examines each of these areas, summarizing the limitations of current research and recommending specific developments that are needed. As these areas are addressed it becomes easier to optimize one or more of the emerging continuous manufacturing technologies that will in turn allow widespread application of bulk nanostructured metals.

Invited report**Research and Development of Nanomaterials and Nanotechnologies in Belarus**Piotr Vitiaz^{a,1}, Vladimir Urbanovich^{b,2}

^a Head of the Belarussian State Program on Nanomaterials
and Nanotechnologies, National Academy of Sciences of Belarus,
66 Nezavisimosti Av., Minsk 220072, Belarus

^b Joint Institute of Solid State and Semiconductors Physics of the NASB,
19 P.Brovki St., Minsk 220072, Belarus

¹ vitiaz@presidium.bas-net.by, ² urban@ifftp.bas-net.by

Among the priorities of scientific research and developments in the Republic of Belarus the creation of new materials and the technologies of their production, the most important of which are nanomaterials is of special important. The research in this direction in Belarus has been already conducted for quite a long time, but nowadays it parted out into a separate State Complex Program, where Institutes of the NASB, high education establishments as well as organizations from various branches of industry participate.

In the report the results of Belarusian scientists in the sphere of theory, experiment, and practical application of nanomaterials and technologies of their production have been considered. This direction is one of priorities in Belarus [1].

The themes of the researches include development of new principles and models for nanoobjects description, methods of research and diagnostics of systems with small-sized structuring: synthesis and research of new nanosized materials (carbon, refractory, composition in the form of ceramics, polymer, layered and membranous structures) of different purpose, development of devices and systems based on the ones.

Great attention is given to the research of the fundamental mechanisms of formation of nanotubes, powder particles, fullerenes, and processes of structure organization of membranous, layered, bulk nanomaterials of different classes in the conditions of high-energy effect (such as arch discharge, laser radiation, high dynamic and static pressures, mechanical alloying, and obtained by the way of electrochemical precipitation, plasmochemical and self-propagating high-temperature synthesis), regularities of nanodevices and systems, based on nanostructured silicon and alumina, operation; also organic and inorganic molecules.

In the report new methods of obtaining and diagnostics of nanomaterials, their characteristics and spheres of application are considered. The results of research in the sphere of device-equipping Atomic Force Microscope for exact interpretation and data-processing of the results obtained for different nanostructured objects are considered [2].

The properties of composition nanomaterials as well as traditional composites with additions of nanostructured carbon elements are considered.

The examples of application of different classes of nanomaterials in different devices perspective for usage in industry are provided. They are nanoelectronic devices based on the porous silicon, carbon nanotubes and alumina, appliances based on Hall's sensors. In Belarusian State University of Informatics and Radioelectronics experimental samples of microdisplay based on nanostructured silicon had been designed and manufactured for the first time in the world. The use of nanostructures in various sensors seems to be very perspective [3].

Innovational way of development of the Republic of Belarus also includes commercialization of the scientific research. Therefore one of the most actual issues for now is market for nanomaterials and nanotechnologies creation, which has not been formed yet in Belarus. The sooner this problem is solved, the faster the system of scientific results utilization in industry will be created.

- [1] P.A. Vitiaz, V.S. Urbanovich, J. Science and Innovations. 7 (2006) 14 (*in Russian*)
- [2] Methodological aspects of Scanning Probe Microscopy, Proceeding of 7th Int. Seminar, Minsk, Institute of Heat- and Mass-Transfer, (2006) 268 p. (*in Russian*)
- [3] P.A. Vitiaz, V.G. Gorobtsov, V.A. Labunov, Nanopages, 3 (2006) 8

Oral report**High Strength Nanostructured TiNi-Based Shape Memory Alloys and
Their Applications**Vladimir Pushin^{a,1}, Ruslan Valiev^{b,2}^a Institute of Metal Physics, Ural Division RAS, 18 S. Kovalevskaya St., Ekaterinburg 620041, Russia^b Institute of Physics of Advanced Materials, Ufa State Aviation Technical University,
12 K. Marx St., Ufa 450000, Russia¹ pushin@imp.uran.ru, ² RZValiev@mail.rb.ru

Our recent complex investigations show that mechanical characteristics of TiNi-based shape memory alloys can be enhanced by using methods of severe plastic deformations (namely high pressure torsion, HPT and equal angular pressing, ECAP). The ECAP-technique allows receiving the bulk nanostructured cylindrical TiNi specimens, 20 mm in diameter and 100 mm in length at different temperatures and strain values (up to a true strain of about 5-10). The long nanostructured rods and wires, up to 0.2 mm in diameter, can be obtained by combination of ECAP and repeated plastic deformation (RPD) with intermediate annealing at temperatures reduced from 700°C down to room temperature. Transmission high resolution electron microscopy X-ray diffraction, measurements of electrical resistivity and magnetic susceptibility were used for investigation of the structural states and phase composition of alloys, their thermostability on annealing and cooling, including in situ experiments for study of thermoelastic martensitic transformations. It was shown that these nanostructured TiNi-based alloys exhibit high strength and ductility, thermoelastic martensitic transformations and related shape memory effects and superelasticity. It is found that unique feature of thermoelastic martensitic transformations in the investigated TiNi-based crystalline alloys is their relatively weak sensibility to different micro-, submicro- and nanostructured states. Thus, TiNi-based alloys have remarkable microstructural states and mechanical properties and there are possibilities for applying them as high strength and ductile shape memory materials. New potential medical and engineering applications of nanostructured TiNi-based alloys for structural and functional using are considered.

Oral report**Commercialization of Nanocrystalline Two-Phase Titanium Alloys Produced by Severe Plastic Deformation**

Gennady Salishchev^{a,1}, Sergey Zherebtsov^{a,2}, Svetlana Malysheva^{a,3},
Oleg Valiakhmetov^{a,4}, Rafail Galeev^{a,5}, Anatoliy Smyslov^{b,6}, Edward Saphin^{b,7},
Sergey Pavlinich^{c,8} and Nailya Izmaylova^{c,9}

^a Institute for Metals Superplasticity Problems of Russian Academy of Sciences,
39 Khalturin St., Ufa 450001, Russia

^b Ufa State Aviation Technical University, 12 K. Marx St., Ufa 450000, Russia

^c JSC Ufa Engine Industrial Association, 2 Ferin St., Ufa 450039, Russia

¹ gensal@anrb.ru, ² ser_zh@imsp.da.ru, ³ svufa@imsp.da.ru, ⁴ olegv@imsp.da.ru, ⁵ galeev@imsp.da.ru, ⁶ umpo@umpo.ru,
⁷ umpo@umpo.ru, ⁸ ion_usatu@mail.rb.ru, ⁹ ion_usatu@mail.rb.ru

Microcrystalline processing of titanium alloys to a nanocrystalline (NC) scale provides increasing their strength characteristics, fatigue limit and wear resistance and also decreasing temperature of superplasticity. Due to such changes in their mechanical properties titanium alloys become advanced for commercial implementation in aero-space industry. The present work considers the results of investigations of structure and mechanical properties for NC titanium alloys and makes their analysis in terms of their practical application. Methods of equal channel angular pressing, multi-step forging and rolling have been used for NC structure processing. Semi-finished products of pilot-industrial dimensions such as plates, rings and sheets have been manufactured from titanium alloys using these methods for evaluating their commercial implementation in aero-space industry. Pilot parts from two-phase titanium alloys- billets up to 150 mm in diameter and 230 mm in length, rings up to 300 mm in diameter and 80 mm in thickness, and sheets, 500 mm in width and 1000 mm in length – have been produced. They had a homogeneous microstructure with grain size less than 0.5 μm and uniform mechanical properties in different sections of parts with excellent properties of strength, fatigue limit and superplasticity.

The application of nanomaterials is distinguished by a number of features attributed to temperatures of their exploitation and possible impact loads. A decrease in high temperature properties is naturally expected at high temperatures. In this connection, the application of NC titanium alloys is possible only in cold part of aircraft engines. Alloys in NC condition (after NC structure processing) are characterized by lower ductility that limits their application in components imposed to impact loads. It has been shown that a modified layer applied to the surface of a blade by means of ion implantation and spraying of protective coating allows increasing fatigue limit and wear resistance of articles processed from these materials.

Processing advantages of practical application of NC titanium alloys for producing aero-space articles have been evaluated using methods of superplastic forging and sheet forming. Unlike the conventional processes the production of semi-products from NC titanium alloys is performed in the temperature range several hundreds degrees below that of usually used for producing such ones. This temperature range is rather attractive in terms of developing resource saving technologies. The reduction of deformation temperature by 200-300°C, as compared with the conventionally used temperatures, results in the considerable savings of electric power, furnace consumables, practically eliminates oxidation, and, consequently, expenditures on the elimination of oxidized/alpha layer and the related ecological activities, makes it possible to increase tool life and to decrease machining costs. It can be expected that attained savings are significantly higher than the necessary expenses/the inevitable costs of increasing strains during formation of nanostructured states.

The work was supported by Russian Foundation of Basic Researches, grants RFBR № 05-08-65396 and Program of Presidium RAS P-08.

Oral report

Bioengineering and Machine-Building Application of Carbon Nanocomposite

Georgy M. Volkov

Moscow State Technical University MAMI,
38 B.Semenovskaya St., Moscow 105839, Russia
recom@list.ru

The unique properties of carbon nanocomposite and opportunity of the manufacture of the real-sized articles in industrial scales create prerequisite for its application in advanced machinery and medical technics.

Carbon nanocomposite is provided durability of a locking element of heart artificial valves, which should maintain about 40 mln double impacts per one year in chemically active environment of native blood. The resource of non-failure operation of nanocomposite in the given conditions is comparable with life expectancy of the man with a 5-times durability reserve.

Application of carbon nanocomposite allowed improving engineering and economical index of thermonuclear reactors Tokamak of a series T-4, T-3M and T-7 essentially. The carbon nanocomposite diafragma blocked a working body of reactors from hydrogenous plasma with temperature about 100 mln degree in 8000 cycles without destruction. It allowed to reduce 3 times losses of capacity of a plasma cord, to increase 5 times quantity of useful pulses and to lower 20 times intensity of rigid X-ray radiations of reactors.

Carbon nanocomposite serves as an effective antifrictional material for face seals working at high circumferential speeds of rotation and high parameters of working environment in energy-saturated friction units of advanced machinery. Carbon nanocomposite surpasses about 5 times traditional materials in friction coefficient. It does not lose its mechanical properties up to 2000°C.

Antifrictional element of gasodynamical bearings made from carbon nanocomposite has't the traces of deterioration after 2000 cycles start-stop at the minimal value of starting friction moment. It works well practically with anyone counterbody. Their important advantage is absence of weldability with a metal surface of the shaft at a short-time contact in case of emergency.

Sphere of realization of properties carbon nanocomposite is much more widely considered directions of its technical application.

Oral report

Numerical Simulation of ECAP at Macro and Meso Level

Andrey Smolyakov^{a,1}, Vyacheslav Soloviev^{a,2}, Kirill Tsiberev^{a,3},
Alexander Korshunov^{a,4}, Marat Abdullin^{a,1}, Nariman Enikeev^{b,c,5}

^a Russian Federal Nuclear Centre, All-Russian Research Institute for Experimental Physics (VNIIEF),
37 Mir St., Sarov 607190, Russia

^b Institute of Physics of Advanced Materials, Ufa State Aviation Technical University,
12 K. Marx St., Ufa 450000, Russia

^c Institute of Mechanics Ufa Science Centre, RAS, 12 K.Marx St., Ufa 450000, Russia

¹smolakov@rol.ru, ²soloviev@vniief.ru, ³tsiberev@vniief.ru, ⁴korshun1@astra.vniief.ru, ⁵carabus@mail.rb.ru

A variation-difference method of solving continuum mechanics equations in combination with the visco-plastic self-consistent model was put forward to perform numerical simulation of equal-channel angular extrusion (ECAE). The developed DRACON code (VNIIEF, Russia) together VPSC code (LANL, USA) allowed carrying out simulations of ECAE at macro- and meso- level, showing good agreement between experimental and calculated data.

Oral report**Development of the Combined Technology for Producing Long – Sized Nanostructured Ti Semi-Products for Structural Application**

Vladimir V. Latysh^{a,1}, Valery M. Polovnikov^{b,2}, Irek V. Kandarov^{a,3},
Gulnaz H. Salimgareeva^{a,4}, Ruslan Z. Valiev^{a,5} and Terry Lowe^{c,6}

^a Ufa State Aviation Technical University, 12 K. Marx St., Ufa 450000, Russia

^b Innovation Scientific Technical Center “Iskra”, 81 Pushkin St., Ufa 450000, Russia

^c “Mettalicum” LLC, New Mexico, USA

¹ latysh-vv@mail.ru, ² polovnikov-vm@mail.ru, ³ IV-Kandarov@mail.ru,

⁴ sadikova_gh@list.ru, ⁵ rzvaliev@mail.rb.ru, ⁶ tlowe@metallikum.com

Despite the existing possibility to produce bulk nanostructured billets and obtain required physical and mechanical properties in Ti, production of such billets still remains a problem [1].

An industrial production demanded a new constructional decision unlike the known laboratory based methods. This is connected with a necessity to reduce the volume of waste material and to enhance constancy of properties, processing speed and low costs of products in the conditions of mass production.

Therefore, our investigations were conducted mainly to improve techniques for processing Ti, including regimes of strain and inter-op thermal processing, to develop die-sets, lubricants, efficient routes of processing, etc.

The investigation results are meant for developing process designs and industrial process for producing long-sized Ti (CP Ti Grade2 and CP Ti Grade4) nanosemi-products with constant high properties.

The given scheme presents a combination of deformation methods: severe plastic deformation (including a stage of preliminary structure preparation) and the subsequent hardening operations.

The first stage (SPD) provides a formation of UFG structure in Ti with grain/subgrain size \cong 500 nm. The feature of the proposed SPD scheme is a combination of operations on forging and ECAP [1]. Such a decision made it possible to reduce the number of ECAP passes up to 2-4, instead of conventionally applied 8, to increase structure homogeneity, to decrease deficient zones at the ends and surface of a billet. As a result of such processing, for example there were achieved the following mechanical properties in CP Ti Grade2: $\sigma_B=740$ MPa; $\sigma_{0,2}=680$ MPa; $\delta = 18\%$, the strength of which is, for example, about 1,5 times higher than that of the initial Ti.

The proposed method of combined deformation on the stage of SPD provided ductility in a metal, necessary for carrying out further deformation processing and for producing long-sized rods ($L=3000$ mm) with record mechanical properties.

For this purpose the obtained at SPD stage billets were subjected to a subsequent strengthening treatment that included such main procedures as multi-pass rolling and drawing. Due to the accomplished investigations there were proposed new engineering solutions, which allowed to optimize deformation and temperature treatment regimes [2], the design of a shape forming die-set [3], lubricant and lubricating coatings[4].

The detailed analysis of the various stages of the combined process made it possible to form a rational version of the technology for producing long –sized billets made of CP Ti with length up to 3 m. The testing of the fabricated samples showed that forming of a homogeneous UFG structure in semi-product made of CP Ti Grade2 with the size of alpha grains 100 nm, allowed to obtain the following mechanical characteristics: $\sigma_b \cong 1100$ MPa, $\sigma_{0,2} \cong 1000$ MPa, $\delta \geq 11\%$, and while realization of the technology for producing semi-products made of CP Ti Grade4 with grains size $\cong 150\div 200$ nm: $\sigma_b \cong 1240$ MPa, $\sigma_{0,2} \cong 1150$ MPa, $\delta \geq 12\%$. Spread in the mechanical properties along a billet was less than $\pm 5\%$, at that the technology provided reduction of metal waste, constant quality of a fabricated product, reduction of labor intensity and prime cost of a process.

The fabricated semi-products are meant for producing high-loaded constructions, for example, medical implants and also for application in automobile - and aerotechnics as fixing products.

- [1] Valiev R.Z., Alexandrov I.V. Bulk nanostructured metal materials, Moscow, “Akademkniga” (2007) 398 p.
- [2] RF patent №2285737, MPK C22F 1/48 B21J 5/00. Publ. 20.10.2006
- [3] USA patent №6399215, MPK C22C 14/00, C22F 01/48. Publ. 04.07.2002
- [4] RF patent № 2128095, MKI 6 B21C 25/100. Publ. 27.03.1999
- [5] GB patent № 2135617, MPK⁴ B21 C23/82 23/08

Oral report**Properties of HPT Processed Al Alloys****Maxim Yu. Murashkin^{a,1}, Askar R. Kilmametov^{a,2}**^a Institute of Physics of Advanced Materials, Ufa State Aviation Technical University,

12 K.Marx St., Ufa, 450000, Russia

¹ maxmur@mail.rb.ru, ² ascar@mail.rb.ru

One of the key tasks tackled at the interface between solid-state physics and pressure shaping is the development of novel approaches that would significantly improve the complex of mechanical and operational characteristics of light Al-based alloys widely used in industry. Such approaches as the achievement of high level of mechanical properties in Al alloys by forming ultrafine-grained (UFG) and nanostructured (NS) states, which demonstrate peculiar “non-equilibrium” grain-boundary behavior and tailored phase composition, are exciting much interest in recent years. To date, it is feasible to obtain such structures in metallic materials through realization of various procedures of severe plastic deformation (SPD) [1]. Generally, the procedures of SPD enable to introduce UFG structure with the size of grains/subgrains varying from 1 μm to several tens of nm in bulk Al alloy billets. At that strength, properties of the materials increase by 30-50 %, as compared to coarse-grained analogies subjected to conventional hardening treatments [2-3].

High pressure torsion (HPT) is one of the current SPD techniques allowing the formation of Al alloys with nano-scale grain sizes ($D \leq 100$ nm), as well as effectively controlling their phase composition. In this connection the work gives the results obtained during the investigation of the effect of HPT processing conditions and subsequent thermal treatment on the NS state formation peculiarities as well as on mechanical properties of heat-non-hardenable Al 1570 alloy (Al-5.7Mg-0.32Sc-0.4Mn) and heat-hardenable 6061 alloy (Al-0.9Mg-0.7Si).

Initial alloy billets having the diameter of 20 mm and 1 mm in thickness were subjected to HPT processing under the pressure equal to 6 GPa till the value of shear strain made up ~ 600 [4]. Before HPT, 1570 alloy billets were annealed at 385°C, and the 6061 alloy billets were solution-treated (ST) at 530°C for 1h and quenched into water. Processing of 1570 alloy by HPT was conducted at ambient temperature and at temperatures of 100°C and 200°C. The 6061 alloy billets were processed at ambient temperature, and then subjected to additional artificial aging at temperatures ranging 70-160°C for up to 48 hours.

It is established that HPT processing makes the formation of NS states in the alloys under investigation possible. The microstructure characteristics found in the alloys before and after straining by the methods of transmission electron microscopy (TEM) and X-ray diffraction (XRD) analysis are given in Table 1.

Alloy	Treatment	D, nm	D _{XRD} , nm	$\langle \varepsilon^2 \rangle^{1/2}$, %	a, Å
1570	Anneal. at 385°C	≥50 μm	-	0.010 ± 0.001	4.0765 ± 0.0001
	Anneal. + HPT at RT	~ 120	45 ± 6	0.190 ± 0.030	4.0692 ± 0.0003
	Anneal. + HPT at 100°C	~ 150	75 ± 7	0.210 ± 0.030	4.0682 ± 0.0002
	Anneal. + HPT at 200°C	~ 210	86 ± 4	0.040 ± 0.002	4.0685 ± 0.0001
6061	ST at 530°C + aging at 160°C, 12 h. (T6)	≥50 μm	-	0.010 ± 0.008	-
	ST + HPT at RT	~ 100	65 ± 3	0.110 ± 0.020	4.0520 ± 0.0001

ultimate tensile strength in the NS 1570 and 6061 alloys reached 950 and 690 MPa, correspondingly (Table 2). At the same time, the Al alloys demonstrate rather high ductility in nanostructured state.

Alloy	Treatment	YS, MPa	UTS, MPa	El., %
1570	Anneal. at 385°C	257 ± 13	394 ± 11	17.0 ± 1.0
	Anneal. + HPT at RT	905 ± 31	950 ± 35	4.7 ± 0.3
	Anneal. + HPT at 100°C	865 ± 25	890 ± 18	4.0 ± 0.4
	Anneal. + HPT at 200°C	845 ± 33		-
6061	ST + aging at 160°C, 12 h. (T6)	276 ± 14	365 ± 16	14.0 ± 1.0
	ST + HPT at RT	660 ± 21	690 ± 28	5.5 ± 0.3
	ST + HPT at RT + aging at 160°C, 0.5 h.	565 ± 15	585 ± 21	13.5 ± 1.5
	ST + HPT at RT + aging at 70°C, 48 h	665 ± 28	700 ± 25	15.0 ± 2.0

The tensile tests performed established that due to the formation of NS structure, the mechanical strength of the Al alloys under investigation can become 2-2.5 times higher. Thus,

The specified decrease in strength of the NS 1570 alloy billets with increasing HPT processing temperature (Table 2) is consistent with its microstructure peculiarities (Table 1). The absence of ductility established in the alloy

after HPT at 200°C and its insignificant reduction after straining at 100°C is largely conditioned by the segregation of Mg atoms and/or the particles of secondary β-phase (Al₃Mg₂) along the boundaries of the forming grains due to the processes of dynamic aging taking place in the material at strain.

It is shown that the ductility of NS 6061 alloy after HPT can be increased by application of additional thermal treatment, namely artificial aging (Table 2). It is established that effective aging temperature, at which the growth of the NS alloy ductility is not attended by its weakening, is lower than the temperature of aging used during conventional strengthening heat treatment (T6) by ~ 100°C.

- [1] R.Z. Valiev *et al.*, JOM April (2006) 33
- [2] M.V. Markushev, M.Yu. Murashkin, in Y.T. Zhu *et al.* (Eds.), TMS Symposium: Ultra-fine Grained Materials II, TMS, Warrendale, PA, USA, (2002) 371
- [3] H.J. Roven *et al.*, Mat. Sci. Eng. A 410-411 (2005) 426
- [4] Vorhauer, R. Pippan, Scripta Mater. 51 (2004) 921

Oral report

**Processing Properties of Nano- and
Submicrocrystalline Ti-6Al-4V Titanium Alloy**

Ramil Ya. Lutfullin^{a,1}, Alexey A. Kruglov^a, Rinat V. Safiullin^a,

Minnaul Kh. Mukhametrakhimov^a and Oleg A. Rudenko^a

^a Institute for Metals Superplasticity Problems, RAS, 39 Khalturin St., Ufa 450001, Russia

¹ lutram@anrb.ru

Method of superplastic forming (SPF), pressure welding (PW) under superplastic conditions and the integral method combining superplastic forming and pressure welding (SPF/PW) are very promising. But their application for processing of titanium alloys is restricted by low life of die tooling and high labor intensity at removing α -case from the surface of processed products. The successful overcoming of all these drawbacks is possible due to the application of decreased processing temperatures, in particular, 760°C [1] instead of 900°C. However, the decrease in temperature within the above identified range and even below it becomes possible only in case of application of a physical effect of “low temperature superplasticity” [2], which is connected with grain refinement from micrometer to a nanometer sizes.

In this respect let us consider the results of full-scale experiments performed in IMSP RAS with bulk nano-, submicrocrystalline and microcrystalline materials.

The experimental studies were performed using processed sheets out of titanium alloy Ti-6Al-4V in three initial states: State A corresponding to the mean grain size 0.2 μm , State B – to the mean grain size 0.4 μm and State C – to the mean grain size 2 μm .

The aim of these studies was to reveal the influence of the mean grain size within the range from 0.2 to 2 μm on the feasibility of joining by pressure welding and the formability of processed sheets under SP deformation conditions.

The decrease in the mean grain size of Ti-6Al-4V alloy sheets from 0.4 to 0.2 μm allows the decrease of the lower temperature of SPF from 750°C to 600°C and the fabrication of cylindrical samples qualitative in their shape, bending radius and thickness distribution.

Due to joining of processed sheets under normal pressure corresponding to low temperature superplasticity flow stress one can decrease the temperature of solid state welding [3]. The sound solid state joint (SSJ) for Ti-6Al-4V alloy in State A was obtained at 600°C (*Table*). The solid state joint with similar shear strength was obtained in submicrocrystalline alloy (State B) at the temperature 700°C (*Table*).

Table. Shear strength of SSJ at room temperature

Initial state of Ti-6Al-4V alloy sheets; Temperature of pressure welding, °C	Shear strength of SSJ (τ), MPa
State B; 700	593,7
State B; 650	215,0
State A; 650	599,1
State A; 600	596,5

Thus, the decrease in the mean grain size to nano- and submicrocrystalline sizes allows the application of the effect of “low temperature superplasticity” for processing hollow components out of Ti-6Al-4V alloy by combining superplastic forming (SPF) and pressure welding (PW).

- [1] P.N. Comley, Materials Science Forum, Trans Tech Publications, 447-448 (2004) 233
- [2] R.Z. Valiev, O. A. Kaibyshev, R.I. Kuznetsov, R.Sh. Musalimov, N.K. Tsenev, Doklady AN SSSR, 4 (1988), 864. (*in Russian*)
- [3] R.Ya. Lutfullin, O.A. Kaibyshev, O.R. Valiakhmetov, M.Kh. Mukhametrakhimov, R.V. Safiullin, R.R. Mulyukov, J. of Advanced Materials, 4 (2003) 326

Guest key-note report**In the Search for New Functional Nanomaterials**Yu.D.Tretyakov¹, E.A.Goodilin

Faculty of Materials Science, Chemistry Faculty of Lomonosov Moscow State University

Lenin Hills, 119991 Moscow Russia

¹yudt@inorg.chem.msu.ru

Modern design of advanced nanomaterials is aimed to the search for the ways of restriction of freedom degrees in open complex systems used for preparation of nanoobjects. It is important to consider the following fundamental parameters characterizing the nanosystems: typical size of the constituent elements (1, 10 or 100 nm), dimensionality (0D, 1D, 2D or 3D), ordering the elements as well as functional properties of initial phases modified by nanosize effects. Thus most of nanomaterials can be generally considered as nanostructured materials possessing an ierarchic structure and polyfunctionality. The key direction of such materials preparation is seemed to be the usage of selforganization and selfassembly principles allowing to achieve prompt and effective formation of nanomaterials. It is tnen important to apply different temlates – intermediate objects controlling interactions on a definite structural level thus forcing the elements of a complex system to selforganize themselves. In the present work a high effectiveness of ionic, molecular, complex templates, surfactants, liotropic liquid crystal matrixes is demonstrated in the production of nanostructured functional materials. It is found that the application of templates plays a very important and universal role in the processes of nanomaterials formation.

Oral report

**Know-How Trade:
Open Forum for Trading Science and Technology Ideas and Solutions,
Know-How, Patents, and Software**

Anatoli Korkin

Nano & Giga Solutions, Gilbert, Arizona, 85296, USA

korkin@nanoandgia.com

The purpose of the business is to develop a world wide internet forum for trading science and technology ideas and solutions, know-how, patents, and software. The forum will be developed to provide an opportunity for small business, intellectual property (IP) owners and investors to acquire due diligence and development assistance such as IP analysis, (development and protection) and technical due diligence assistance to the investors, HR and financial needs assistance to the inventors and manufacturing assistance for small and large corporations. The final goal of the business is to develop a universal tool for low risk, low cost and time efficient process for small to large business development from idea to a mature business with an opportunity to select independent service providers.

With ever increasing pressure to introduce the latest technologies into the market place, a world wide Internet Forum for trading technological and intellectual properties is required. Developing a business model around this unique market need enables transactions at all levels of intellectual property, science, and technology to go forward. For example, this forum would present many development and technology ideas and solutions, know-how, process design, trade practices, patents, software, etc. The market need for this forum is enormous and growing because of the extremely short period of time from invention to market place. Traditional methods and techniques for selling and acquiring intellectual property, science, and technology has several problems, such as time, consumption of management resources, contact of willing parties, and the like. The growing market need for an internet based know-how trade forum is demonstrated by the number of efforts in this space, failed and successful. A successful Internet Forum that specializes in intellectual property, science, and technology can potentially eliminate or at least greatly reduce many problems involved with the transactions of intellectual property, science, and technology. By having an identifiable forum with world wide recognition with minimal cost that offers standard practices, discrete initial inquires and interaction between sellers and buyers, will allow willing buyers and sellers to be able to transact business with lower transaction costs. Additionally, searchable databases of available ideas and/or solutions that are directly accessible for low cost and easy to use to

technical experts, managers and decision makers will lower the barrier to enter the market and result in a large worldwide usage. In the future internet base trade of IP will likely become the major business model for know-how transfer.

List of Authors

A

Abdullin M.F.	204, 308
Ahmadabadi M.N.	221
Aifantis E.C.	87, 251
Akbari Mousavi S.A.A.	205, 245
Akbarzadeh A.	279
Alexandrov I.V.	119, 178, 179, 198, 204, 242, 244, 296
Amirhanova F.A.	152
Amirhanova N.A.	120, 121, 142, 152, 166
Amouyal Y.	208
Andrievskaya N.F.	176
Andrievski R.A.	17, 28
Anisimov A.G.	40
Antonova O.A.	124
Astafiev V.V.	124
Azhazha V.M.	176

B

Bachurin D.	246
Balabanova N.V.	122
Barber R.E.	12
Bardinova (Nikitina) M.A.	115
Baretzky B.	193
Barkai I.	35
Barykin N.P.	50
Baryshnikov M.P.	134
Bastarache E.N.	126
Baushev N.G.	60
Beliakov N.	81
Belousov M.	143
Belousov N.N.	39
Bengus V.Z.	172, 280
Bernstorff S.	46
Beyerlein I.	106, 204
Beygelzimer Y.	11, 181
Blinov S.V.	53
Bobylev S.	190
Böckmann R.	133
Bonarski J.	242
Botkin A.V.	291
Botta Filho W.J.	23
Brailovski V.	143, 282
Brodova I.G.	124
Budilov I.	244, 299

C

Chanchetti L.F.	23
-----------------	----

Chandrasekar S.	232
Charlot F.	103
Chembarisova R.	198
Chernavina A.	61
Chernov V.	248
Chertovskikh S.V.	64
Chokshi A.H.	111
Choo H.	172
Chukin M.V.	134
Cieslar M.	194
Compton W.D.	232
Csach K.	280

D

Danilenko V.N.	51
Davidenko A.	181, 286
Demers V.	282
Ditenberg I.	248
Divinski S.	208,
Dobatkin S.V.	19, 22, 31, 80, 126, 127, 143, 167, 168, 282
Dobatkina T.V.	167, 168
Drobyshev V.	81
Druzycka-Wienczek A.	128
Dudarev E.F.	275
Dudina D.V.	40
Dudova N.	272
Dzyabchenko A.	249

E

Efros B.	89, 157, 181
Efros N.	89, 157
Eggeler G.	133, 227
Emaleeva D.G.	134
Emaletdinov A.K.	253
Emel'chenko G.A.	70
Enikeev N.A.	140, 204, 308
Ermolenko A.	298
Estrin Y.	167, 168, 208

F

Faizova S.N.	122, 129, 286
Farzetdinova R.M.	131
Fecht H.-J.	104, 184
Ferrante M.	79
Fesenko V.A.	33, 44
Figueiredo R.B.	49, 212
Fodor A.	35
Frotscher M.	133
Furukawa M.	20

G

Gabor P.	225
Galeyev R.	305
Galimov A.K.	270
Ganeev A.V.	179, 296
Gimazov A.A.	254
Gladkovskii S.	89
Glezer A.	31
Göken M.	210, 214, 224, 236
Golosov E.	76
Gordopolov Y.A.	70
Gornostyrev Yu.N.	259
Grabovetskaya G.P.	76, 139
Grekhov M.M.	33, 44
Grosman F.	128
Gubicza J.	35
Gumbsch P.	246
Gun G.C.	134
Gunderov D.V.	59, 60, 100, 238
Gür C.H.	218
Gurtovaya (Trubitsyna) I.	13, 143, 216
Gutkin M.	190

H

Hartwig K.T.	12, 25, 156, 180
Hashimoto S.	112
Herzig Ch.	208
Hietschold M.	230
Hisaev R.N.	115
Hockauf M.	230
Hohenwarter A.	136
Höppel H.W.	210, 214, 224, 236
Horita Z.	20, 26
Hüller M.	236

I

Im Y.-T.	220
Imayev M.	137
Inaekyan K.	216, 282
Isaenkova M.G.	33, 44
Iskandarov A.M.	256
Islamgaliev R.K.	115, 150, 163, 234
Ivanisenko Y.	104, 184
Ivanov A.M.	43, 59
Ivanov K.V.	76, 139
Ivanov M.B.	42, 76
Ivanov V.	71
Ivchenko V.	157

Izmaylova N. 305

J

Jakushina E. 154
Jayaganthan R. 83
Jin Y.G. 220
Jinbo N. 55
Jorge Jr. A.M. 23

K

Kad B. 113
Kamaeva L.V. 48
Kandarov I.V. 258, 286, 309
Karaman I. 106, 223, 284
Karkin I.I. 259
Karkina L.E. 259
Karnthaler H.P. 57, 169
Karpuz P. 218
Kasatkin D. 202
Kashin O.A. 275
Kawasaki M. 49, 110
Kazykhanov V. 140
Kecskes L.J. 12, 25, 156, 180
Kerber M.B. 46, 228
Kestenbach H.-J. 23
Khaidarov R.R. 142
Khazgaliev R. 137
Khlebova N. 81
Khmelevskaya I. 143, 216, 282
Khomskaya I. 66
Kilmametov A.R. 311
Kim H.S. 196
King A.H. 232
Klemm V. 260
Klimanek P. 260
Kodjaspirov G.E. 22
Kogtenkova O.A. 193
Kolesnikova A.L. 260
Kolobov Yu.R. 42, 76, 139, 275
Kommel L. 262
Korkin A. 316
Korotaev A. 248
Korotitskiy A. 143, 216
Korshunov A.I. 145, 204, 242, 308
Korshunov L. 159
Korzniikov A.V. 146, 159, 228, 266
Korzniikova E. 228
Korzniikova G.F. 146
Kovarik T. 194

Kozlenkova N.	81
Krallics G.	35
Krasilnikov N.A.	115, 148, 149
Kravchenko T.	145
Krüger L.	230
Kruglov A.A.	264, 313
Kucukomeroglu T.	284
Kulyasova O.	234
Kuprin C.	230
Kupriyanov A.A.	174
Kuranova N.N.	100
Kurmanaeva L.R.	150
Kurzydłowski K.J.	128
Kutnyakova J.B.	152
Kuziak R.	128
Kuzminyh A.A.	50, 291

L

Lad'yanov V.	48
Langdon T.G	5, 20, 49, 99, 110, 212
Lapovok R.	15, 168
Latysh V.V.	114, 258, 286, 309
Lebedev Yu.A.	51
Leiva D.R.	23
LeMoulec A.	103
Lewandowska M.	251
Li H.	172
Li S.	92, 204
Liaw P.K.	172
Litovchenko I.Yu.	85, 266
Litvinov S.D.	289
Liu J.Q.	296
Liu M.	96
Loladze L.	89
Lomayeva S.	71
Lomovsky O.I.	40
Lopatin N.V.	50
Lowe T.C.	36, 301, 309
Lukaschuk Yu.	299
Lukin E.S.	43, 59
Lukyanov A.V.	100, 238
Lutfullin R.Ya.	264, 270, 313

M

Maier H.J.	223, 225
Mali V.I.	40
Malysheva S.	154, 260, 305
Mann J.B.	232
Manohin S.S.	42
Markovchev V.A.	115

Markushev M.V.	268
Martin M.	113
Masimov M.	260
Mastoori M.	205, 245
Mathaudhu S.N.	12, 25, 156, 180
May J.	210, 225
Mazilkin A.A.	193
Mazitov R.M.	51
Meilikhov E.Z.	131
Memetov N.R.	53
Metlov L.	157
Meyer L.W.	230
Meyers M.	113
Mimaki T.	55
Mingler B.	169, 234
Minkov A.	104, 234
Mishra A.	113
Miskuf J.	280
Miyamoto H.	55
Mizera J.	165
Molokanov V.V.	48
Morozov A.	145
Moscoso W.	232
Motylenko M.	260
Mukhametrakhimov M.Kh.	270, 313
Mukhtarov Sh.	272
Mulyukov R.R.	91
Murashkin M.Yu.	96, 163, 268, 311
Murzaev R.T.	273
Murzinova M.A.	63, 219

N

Naidenkin E.N.	139
Naidenkin E.V.	127, 275
Naimark O.	186
Naumenko I.G.	70
Nazarov A.A.	200, 204, 256, 273
Nedjad S.H.	221
Neuking K.	133
Niendorf Th.	223
Nikitina (Bardinova) M.A.	115
Nikitina N.I.	167, 168
Noskova N.	159
Novy Z.	194
Nowakowski P.	165
Nurgaleeva V.V.	161
Nurislamova G.V.	115, 163

O

O'Donnell R.	15
--------------	----

Odessky P.D.	127
Olejni L.	21, 165
Orlov D.	11
Ota K.	103
Ovid'ko I.	190

P

Pakiela Z.	165
Panigrahi S.K.	83
Pantsyrny V.	81
Patselov A.	89
Pavlinich S.	305
Perevezentsev V.N.	126, 202, 277
Perevoshchikova N.	61
Perlovich, Y.A.	33, 44
Perov N.	31
Peterlechner M.	57
Petrov P.P.	59
Petrova N.D.	59
Pikalov A.I.	174
Pilyugin V.	89
Pinzhin Y.	85, 248
Pippan R.	136
Pirgazi H.	279
Platonov A.A.	59
Plotnikova M.	31
Pochivalova G.P.	275
Podolskiy A.V.	172, 280
Polovnikov V.M.	258, 309
Polyakov A.V.	60
Polyakov L.	145
Popov A.G.	100
Popov M.V.	167, 168
Prados E.F.	79
Prasad M.J.N.V.	111
Prokofiev E.A.	100, 143, 238
Prokoshkin S.	13, 61, 143, 216, 282
Prudnikov A.	170
Pupynin A.	277
Purcek G.	284
Pushin V.G.	100, 238, 304

R

Raab G.I.	36, 59, 60, 126, 127, 129, 214, 241
Rabkin E.	208
Ramesh K.T.	180
Ratochka I.V.	139
Razyapova A.F.	166
Reshetov A.	11
Ribarik G.	46

Ribbe J.	208
Rokhlin L.L.	167, 168
Romanov A.E.	77, 260
Rösner H.	9
Rosochowski A.	21
Rostova T.D.	126
Roven H. J.	96
Rudenko O.A.	264, 313
Rudicheva T.Yu.	176
Rudskoy A.I.	22
Rybin V.V.	18, 98
Ryklina E.	61

S

Safiullin A.R.	63
Safiullin R.V.	63, 264, 313
Saitova L.R.	214, 280
Saldana C.	232
Salimgareeva G.Kh.	114, 169, 258, 309
Salishchev G.	154, 219, 305
Saphin E.	305
Sarafanov G.	202
Saray O.	284
Sarkeeva E.A.	286
Sauvage X.	192
Saxl I.	108
Schafler E.	46, 188, 228
Schegoleva N.N.	100
Schmitz G.	208
Schulz U.	227
Sechnoy A.I.	289
Seefeldt M.	98
Semenov V.A.	51
Semenov V.S.	64
Semenova I.P.	114, 161, 169, 214, 280
Serebryany V.N.	168
Setman D.	228
Shagalina S.V.	127
Shahab A.R.	205, 245
Shalayev R.	170
Shalimova A.	31
Shankar M.R.	232
Sharafutdinov A.	148, 178
Shayahmetov A.F.	291
Shcherbakov A.	288
Sheinerman A.	190
Shevchenko N.V.	85, 266
Shikov A.	81
Shirinkina I.G.	124
Shorokhov E.V.	66, 124
Shuster L.Sh.	64

Şimşir C.	218
Sitdikov V.D.	198, 242
Skiba N.	190
Sklenicka V.	108
Smirnov S.N.	172, 280
Smolyakov A.A.	145, 204, 308
Smyslov A.	305
Soloviev V.P.	204, 308
Sordi V.L.	79
Soshnikova E.P.	60
Sterkhova I.V.	48
Stolyarov V.V.	13, 100, 143, 216
Straumal B.B.	193
Sudakova T.V.	289
Suś-Ryszkowska M.	128
Svoboda M.	108
Synkov A.	11
Synkov S.	11
Syromyatnikova A.S.	59
Syugaev A.	71

T

Tabachnikova E.D.	172, 280
Tak K.-G.	227
Tarkowski L.	242
Tarytina I.E.	167, 168
Tatyanin E.	143, 282
Thadhani N.	113
Theissmann R.	104
Tikhonovsky M.A.	174, 176
Timofeev V.N.	167, 168
Timokhina I.	15
Titorov D.	293
Tkachev A.G.	53
Tolmachev I.D.	176
Tomé C.N.	106
Topic I.	224
Tretyakov Y.	315
Trubitsyna (Gurtovaya) I.	13, 143, 216
Trumble K.P.	232
Tsenev A.N.	118
Tsenev N.K.	118
Tsiberev K.V.	308
Turlakov D.A.	53
Tyumentsev A.N.	85, 248, 266

U

Uksusnikov A.	238
Ulyanov A.	170
Ungár T.	119, 188

Urbanovich V. 17, 302
Utyashev F.Z. 241

V

Valiakhmetov O. 305
Valiev R.Z. 6, 36, 104, 180, 193, 214, 304, 309
Valitov V. 68, 272
Van Boxel S. 98
Van Houtte P. 98
Varyukhin V. 11, 89, 157, 170
Vaughan G. 103
Vedernikova I. 145
Velikodny A.N. 176
Veretennikov V.A. 70
Vershinina T.N. 42
Vinogradov A. 112
Vitiaz P. 302
Vlcek J. 236
Volkov G.M. 295, 307
Vorobieva A. 81
Vychigzhanina L. 178

W

Waitz Th. 57
Wang J.T. 29, 178, 179, 296
Wawer K. 251
Wei Q. 156, 180
Weissmüller J. 74
Weygand D. 246
Wieczorek A.K. 46, 251
Wilde G. 9
Wolff K.-D. 133

X

Xia S. 178, 179
Xu C. 49, 99

Y

Yablonskikh T.I. 124
Yadrovski E.L. 51
Yagodkin Y. 80
Yakovec A. 170
Yang K. 104
Yang P. 232
Yapici G.G. 106
Yavari A.R. 103
Yelsukov E. 71
Yunusova N.F. 115

Z

Zabolotny S.	137
Zakharov V.V.	126
Zavdoveev A.	181
Zayatz S.	71
Zehetbauer M.J.	8, 46, 169, 188, 228, 234, 251
Zhang Y.	296
Zherebtsov S.	219, 305
Zhernakov V.	244, 298, 299
Zhgiliev I.I.	66, 124
Zhilyaev A.P.	94, 254
Zhu Y.T.	36, 184
Zięba P.	193
Zillmann B.	230
Zisman A.A.	98
Zrnik J.	194

